Al-Ni-Re (Aluminum-Nickel-Rhenium)

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The computed phase equilibria of this ternary system by [1999Hua] were reviewed in the update by [2006Rag]. The update presented a tentative liquidus projection, an iso-thermal section at 1040 °C for Ni-rich alloys and a vertical section along the NiAl-Re join. Recently, [2009Gru] investigated this system between 1030 and 700 °C and found a ternary phase in the Al-rich region. Earlier, [2007Sai] had determined a number of tie-lines at 1150 °C.

Binary Systems

The Al-Ni phase diagram [1993Oka] shows five intermediate phases: NiAl₃ ($D0_{11}$, Fe₃C-type orthorhombic), Ni₂Al₃ ($D5_{13}$ -type hexagonal, denoted δ), NiAl (B2, CsCltype cubic, denoted β), Ni₅Al₃ (Ga₃Pt₅-type orthorhombic), and Ni₃Al ($L1_2$, AuCu₃-type cubic, denoted γ'). The Al-Re phase diagram [2008Bal, 2009Gru, 2009Oka] depicts the following intermediate phases: Al₁₂Re (Al₁₂W-type cubic), Al₆Re (Al₆Mn-type orthorhombic), Al₄Re(HT) (monoclinic), Al₄Re(LT) (triclinic), Al₃Re, Al₁₁Re₄ (Al₁₁Mn₄type triclinic), AlRe (B11, CuTi-type tetragonal), and AlRe₂ ($C11_b$, MoSi₂-type tetragonal). The Ni-Re phase diagram is of the simple peritectic type [Massalski2].

Ternary Isothermal Sections

With starting metals of 99.999% Al, 99.98% Ni, and 99.95% Re, [2009Gru] levitation-melted about 10 Al-rich



Fig. 2 Al-Ni-Re partial isothermal section for Al-rich alloys at $870 \ ^{\circ}C \ [2009Gru]$



Fig. 1 Al-Ni-Re partial isothermal section for Al-rich alloys at 1030 °C [2009Gru]



Fig. 3 Al-Ni-Re partial isothermal section for Al-rich alloys at 800 $^{\circ}\text{C}$ [2009Gru]



Fig. 4 Al-Ni-Re partial isothermal section for Al-rich alloys at 700 °C [2009Gru]



Fig. 5 Al-Ni-Re tentative partial isothermal section at 1150 °C [after 2007Sai]

ternary alloys under Ar atm. The alloys were annealed at 1030, 870, 800, and 700 °C for 140, 210, 1313, and 306 h, respectively. The phase equilibria were studied with x-ray

powder diffraction, scanning and transmission electron microscopy, energy dispersive x-ray analysis, and differential thermal analysis at heating/cooling rates of 20 °C per min. The partial isothermal sections constructed by [2009Gru] at 1030, 870, 800, and 700 °C are shown in Fig. 1-4. At 1030 °C (Fig. 1), liquid, Al₄Re(HT), and Ni₂Al₃ are present. No ternary phase was found. Al₄Re(HT) dissolves up to \sim 4.5 at.% Ni. The solubility of Re in Ni₂Al₃ is less than 0.5 at.%. At 870 °C (Fig. 2), the ternary phase X is present around the composition Al_{73 5}Ni_{18 5}Re₈. It has orthorhombic symmetry (*Pbm2* or $Pb2_1m$ or Pbmm), with the lattice parameters of a = 1.0048 nm, b = 1.5423 nm, and c = 0.8367 nm[2009Gru]. It forms peritectically at 888 °C. Al₄Re(LT) dissolves at least 1.5 at.% Ni at 870 °C. The tie-triangle between Al₄Re(HT), Al₄Re(LT) and liquid shown in Fig. 2 is provisional. At 800 °C (Fig. 3), NiAl₃ is present and forms tie-lines with Al₄Re(HT). A narrow tie-triangle (not shown) is expected between Al₆Re, Al₄Re(LT) and liquid. At 700 °C (Fig. 4), only two samples with 90 and 85 at.% Al were investigated by [2009Gru]. The solubility of Ni in $Al_{12}Re$ and Al_6Re is about 1 at.%.

[2007Sai] arc-melted six ternary alloys containing 36.4-81.8 Ni, 9.1-54.5 Al, and 9.1 Re (at.%). The alloys were annealed at 1150 °C for 1000 h and quenched in water. The phase compositions were measured with electron probe microanalyzer. The tie-lines of coexisting phases were listed. Using the tie-line results and the binary phase diagrams as a guide, a tentative isothermal section at 1150 °C is drawn in Fig. 5. No ternary phases were found at 1150 °C. The maximum solubility of Ni and Al in (Re) is 28 and 1.2 at.%, respectively.

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